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# Self-cleaning and air purification performance of Portland cement paste with low dosages of nanodispersed TiO<sub>2</sub> coatings



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# HIGHLIGHTS

• The stable nanodispersed anatase TiO<sub>2</sub> particles are synthesized in hydrosol form.

• The TiO<sub>2</sub> hydrosol-based coating presents a much better self-cleaning performance than that of P25 suspension.

• The TiO<sub>2</sub> hydrosol-based coating shows excellent air purification properties.

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# $A \hspace{0.1in} B \hspace{0.1in} S \hspace{0.1in} T \hspace{0.1in} R \hspace{0.1in} A \hspace{0.1in} C \hspace{0.1in} T$

The strong agglomeration potential and poor re-dispersibility of nano  $TiO_2$  particles in powder form seriously affect the depollution and self-cleaning performances. The nano  $TiO_2$  particles in hydrosol form can be stably dispersed in aqueous solution, but the synthetic temperature influences the crystal pattern, stability, and hydrodynamic diameter of nano  $TiO_2$  particles. In this study, a stable and nano dispersed anatase  $TiO_2$  hydrosol is synthesized by hydrolysis of titanium isopropoxide, and the effects of synthetic temperature on physicochemical properties and stability of  $TiO_2$  hydrosol are discussed. The photocatalytic and self-cleaning performances of synthetic  $TiO_2$  hydrosol coating are characterized by the colour change rate of Rhodamine B and  $NO_x$  conversion UV irradiation. Compared with a commercial nano  $TiO_2$  powder (Degussa P-25), the  $TiO_2$  hydrosol coating with very low dosages shows better photocatalytic and self-cleaning performances under the same experimental conditions.

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# 1. Introduction

Cementitious materials in buildings are directly and continuously exposed to various atmospheric pollutants and microorganisms under different weather conditions. Colour, the main index of aesthetic properties of buildings, must give a pleasant appearance that should give the public an adequate perception of the quality with the low necessity of maintenance. The leading cause of colour change on the surface of cementitious materials is the reduction in initial solar reflectivity, mainly from atmospheric aerosol pollutants such as nitrogen oxides, carbon-based materials (for example, soot), and volatile organic compounds [1,2]. Nano TiO<sub>2</sub>-based coatings have been studied to improve the air purification and the self-cleaning property of cement-

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based materials [3-10]. The nano  $\text{TiO}_2$  powders (for example, Degussa P25) [6,11-13] have been used as concrete coatings to enhance the photocatalytic activity. However, due to the inherent agglomeration and poor re-dispersity of nano  $\text{TiO}_2$  powders in the aqueous system [14], the air-purifying and self-cleaning of coatings made of  $\text{TiO}_2$  nanopowders are impacted obviously. Thus, a relatively high dosage of nano  $\text{TiO}_2$  powders is often used to obtain good depollution performance, which leads to lower transparency of coatings and economic issues.

According to the previous studies [14–20], the photoactivity of nano  $TiO_2$  particles is significantly associated with the crystal pattern, crystalline particle size, and size distribution of nanoparticles in aqueous synthetic nano  $TiO_2$ , which are demonstrated by the different synthesis processes of nano  $TiO_2$ . There are many methods to synthesize nano  $TiO_2$ , but the sol–gel process is widely used because of its advantages including such as high purity, excellent uniformity, the controllable microstructure of product, easier, low temperature, and low cost in processing. The heat treatment of calcination in the sol–gel process is necessary to transform

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amorphous TiO<sub>2</sub> to crystalline when the route is the commonly used polymeric route [14,17,20,21]. In this polymeric route, the solvent is usually alcohol, and the hydrolysis of titanium alkoxide precursors is carefully controlled by using a limited amount of water. However, the solvent is water in the second route [21,22], in which process the produced TiO<sub>2</sub> nanoparticles in hydrosol is crystalline without the step of calcination. Acids are used as peptizators to produce a TiO<sub>2</sub> hydrosol containing nanoparticles with a hydrodynamic diameter in the range of 15 nm to 100 nm. The crystalline TiO<sub>2</sub> nanoparticles are predominantly anatase with brookite and rutile, depending on the used acids [21,23,24].

Moreover, TiO<sub>2</sub> hydrosol [20-22,25-32] has a much larger specific surface area and is much more transparent than conventional TiO<sub>2</sub> nanopowder catalysts, which demonstrated better performance for the photocatalytic degradation of pollutants. As the photodegrading reactions between pollutants and coatings occur on the surface of TiO<sub>2</sub> particles [33], the excellent dispersity of nano TiO<sub>2</sub> particles in a liquid is critical that influences self-cleaning and photocatalytic performances of coatings. The stability of the TiO<sub>2</sub> hydrosol is another factor that needs to be considered in the synthetic process. For large-scale production of TiO<sub>2</sub>-based coatings (for example, roller coating), the long-term stability of TiO<sub>2</sub> hydrosol is required to achieve acceptable repeatability. However, very few studies are available on the effects of synthesis temperature on the stability of TiO<sub>2</sub> nanoparticle dispersion in water.

This study investigates the influences of synthetic temperature in the so-gel process on the dispersion stability of produced  $TiO_2$ hydrosol in water. In a second step, the self-cleaning and air purification performances of  $TiO_2$  hydrosol-based coatings are tested by applying on the hardened cement paste's surface at the age of 7 days.

# 2. Materials and experimental

# 2.1. Materials

Titanium tetra-isopropoxide (TTIP, 97.0%) is purchased from Sigma-Aldrich. Glacial acetic acid (99.6%), and absolute ethanol (99.9%), are purchased from VWR Chemicals. Deionized water (18.2 M $\Omega$ .cm) is used throughout the development process of TiO<sub>2</sub> hydrosol. The nano TiO<sub>2</sub> powder (P25) is purchased from Evonik Industries AG company. The P25 powder contains about 75 wt% of anatase and 25 wt% of rutile, with the primary crystalline particle size of 10 to 50 nm and the specific BET surface area of 50 ± 15 m<sup>2</sup>/g [3,34].

The selected method in this study to produce anatase  $TiO_2$  hydrosols is adapted from studies from Yang et al. [35] and Alphonse et al. [21], but the different synthesis temperatures are applied. The synthesis is as follows: TTIP is dissolved in absolute ethanol with the TTIP/ethanol molar ratio of 2.44, the solutions are stirred for 30 min at different temperatures (50 °C and 40 °C). The obtained solution is added drop by drop at the speed of 0.01 mL/s into a mixture containing acetic acid and deionized water with a molar ratio of 0.175. After that, the suspension is continuously stirred for 48 h at 50 °C or 40 °C by magnetic stirrers and then left to settle for at least 72 h at a room temperature of 20 °C and relative humidity of 60%.

CEM I 52.5 R cement and tap water are used to prepare cement paste samples with the water to cement ratio of 0.4. Two different sizes of samples are prepared for different characterization tests. The paste samples in the size of  $40 \times 40 \times 40$  mm<sup>3</sup> are prepared for testing the surface self-cleaning performance and reflectance property after coating the prepared TiO<sub>2</sub>. The paste samples in the size of  $100 \times 200 \times 5$  mm<sup>3</sup> are prepared for testing the NO<sub>x</sub> degradation performance. After one day curing at the ambient

environment, the paste samples are demoulded and cured in curing chamber (95% R. H. and 20 °C) till further tests. The top surfaces of cement paste samples are polished by SiC sandpapers to obtain relatively smooth surfaces with the roughness in the range of 10  $\mu$ m to 14  $\mu$ m. The synthetic TiO<sub>2</sub> hydrosol and P25 water suspension, which have the same mass content of TiO<sub>2</sub> particles, are coated on the surface of hardened cement paste samples. One cement paste sample is prepared for each coating dosage in the same test.

# 2.2. Methods

# 2.2.1. Characterization

The phase composition of the dried  $TiO_2$  powder sample is investigated by comparing X-ray diffraction (XRD) pattern (Bruker D4 PHASER, Philips, The Netherlands) with a Co tube (40 kV, 40 mA). A typical run used in this test is made with a step size of  $0.02^{\circ}$ /min and a dwell time of 0.5 s. The crystallite size (D<sub>c</sub>), also named the primary particle size [36], of the anatase phase is estimated from the Scherer's equation [24].

$$\mathbf{D}_{\mathbf{c}} = k\lambda/\beta \cos\theta \tag{1}$$

where K = 0.89,  $\lambda$  = 0.154 nm,  $\beta$  is the full-width height maximum in radians, and  $\theta$  is the Bragg's angle.

The particle size distribution and zeta potential of  $TiO_2$  hydrosol are tested by the Zetasizer NanoSeries (Malvern Panalytical) at 25 °C. In these tests, the initial hydrosol is diluted 100 times in distilled water. The  $TiO_2$  hydrosol samples are tested three times by Zetasizer Nano Series using a Dynamic Light scattering process. The Zetasizer Nano Series calculated the zeta potential by determining the electrophoretic mobility and then applying the Herny equation.

The UV–VIS absorbance spectra of  $TiO_2$  hydrosol and P25 suspension samples are measured by the UV–VIS-NIR spectrophotometer (Perkin Elmer Lambda 750) tested range is 250 nm to 800 nm, 1 nm per second. According to the Beer-Lambert law, the sample content should be lower than 0.01 M, and the absorbance should be set below 1.0, in this study, the P25 suspension sample is diluted down to 0.02% and the  $TiO_2$  hydrosol sample is diluted down to 1.0% by distilled water.

To measure the specific surface area and crystal phase pattern of the  $TiO_2$  hydrosol sample, the powder sample is prepared by drying the initial  $TiO_2$  hydrosol in the oven at 105 °C for 24 h. The specific surface area of dried  $TiO_2$  hydrosol is measured by The Brunauer, Emmett, and Teller (BET) method. The pore size distribution from the adsorption isotherm is evaluated by the Barrett-Joyner-Halenda (BJH) interpretation. Nitrogen sorption isotherm experiments are carried out at 77 K temperature by a nitrogen adsorption/desorption device of type TriStar II 3020.

# 2.2.2. Self-cleaning performance of coated cement paste

The self-cleaning performance of the TiO<sub>2</sub> coated cement paste is evaluated by colorimetric analysis of the degradation of the organic dye Rhodamine B (RB) [37]. The coated weights of TiO<sub>2</sub> particle in TiO<sub>2</sub> hydrosol coatings are 0.77 g/m<sup>2</sup>, 1.54 g/m<sup>2</sup>, and 3.08 g/m<sup>2</sup>. The cement paste samples without TiO<sub>2</sub> coating and coated P25 water suspension with the coating dosage of 1.54 g/ m<sup>2</sup> are tested as the reference groups. After coating the TiO<sub>2</sub>, the paste samples are dried in a dark box for 12 h under an ambient temperature of 20 °C. After pre-coated with TiO<sub>2</sub> hydrosol or P25 suspension, the top surfaces of cubic cement paste samples are stained by painting 600 uL of 0.1 mM RB aqueous solution. Then the cement paste samples are also dried in a dark box at 20 °C for 12 h. For each tested cement paste sample, nine points are considered for the colorimetric tests and each point is tested four times. The samples are exposed to a UV lamp (10 ± 0.05 W/m<sup>2</sup>) to simulate UV light in natural outdoor conditions, monitoring the discoloration of the RB on the surface of mortar.

The reflected colour measurements are taken directly on the surface of each point on each sample at different illumination with a spectrometer USB4000 Oceanoptics, which is optimized for the 380–780 nm wavelength range and analyzed mathematically to yield colorimetric quantities like xyz, RGB or L\*a\*b\*. In this study, the percentage of discoloration ( $R_t$ ) is expressed with the coordinate of the dominant colour of dye a\*, a value of the CIE Lab colour space for RB [38–40], according to Eq. (2).

$$R_t(\%) = \frac{a_0^* - a_t^*}{a_0^*} \times 100 \tag{2}$$

Where,  $a_0^*$  the value of  $a^*$  at time 0 before irradiation,  $a_t^*$  its value after t minutes irradiation.

# 2.2.3. Air purification performance of coated cement paste

The air purification experiments [6,7,11,41] of coated cement paste are carried out in a homemade reactor designed following the standard ISO 22197–1. The contents of TiO<sub>2</sub> particles in hydrosol and P25 suspension coatings are 1.54 g/m<sup>2</sup> and 3.08 g/m<sup>2</sup>. The experimental setup, as shown in Fig. 1, consisted of a planar reactor cell, an UVA light source, a chemiluminescent NOx analyzer, and gas supply. The chief operating conditions of the system are as follows: the wavelength of UV light resource is 300 nm to 400 nm, the irradiance flux on the surface of flat cement paste samples is 10 ± 0.05 W/m<sup>2</sup>, the pollutant source concentration is 1.0 ppm, the NO flow, and airflow are 0.06 L/min and 2.94 L/min, and the total gas flow is 3.0 L/min, the relative humidity in the reactor is 50 ± 1%. The amount of NO<sub>x</sub> converted in the reactor is calculated following:

$$NO_{X}Conversion(\%) = \frac{[C_{NOX}]_{in} - [C_{NOX}]_{out}}{[C_{NOX}]_{in}} \times 100$$
(3)

Where  $[C_{NOx}]_{in}$  is the initial concentration [ppm], measured by taking the average value of the first 5 min of the experiment, before turning on the light. The outlet concentration  $[C_{NOx}]_{out}$  is measured by taking the average value of the last 5 min of the irradiation period [ppm].

The amount of NO converted in the reactor is calculated following:

$$NOConversion(\%) = \frac{[C_{NO}]_{in} - [C_{NO}]_{out}}{[C_{NO}]_{in}} \times 100$$
(4)

Where  $[C_{NO}]_{in}$  is the initial concentration [ppm], measured by taking the average value of the first 5 min of the experiment, before turning on the light. The outlet concentration  $[C_{NO}]_{out}$  is measured by taking the average value of the last 5 min of the irradiation period [ppm].

# 3. Results and discussions

#### 3.1. Phase type of TiO<sub>2</sub> in hydrosol

The diffraction angle  $2\theta$  of (101) crystal plane of anatase TiO<sub>2</sub> crystal is 29.4°, the (004) crystal plane is  $44.2^{\circ}$  and the (200) (105) and (204) crystal planes are 56.4°, 63.5°, and 74.3°, respectively [42]. As can be seen in Fig. 2, the TiO<sub>2</sub> particles in hydrosol synthesized at 40 °C and 50 °C shows all the typical diffraction peaks of anatase, indicating the synthesized TiO<sub>2</sub> particles in hydrosol are pure anatase crystals. According to the calculations applying Eq. (1), the crystalline sizes of synthetic TiO<sub>2</sub> particles are 12 nm and 16 nm in hydrosol samples with processing temperatures of 40 °C and 50 °C, respectively. These results prove that the crystal pattern of TiO<sub>2</sub> particles in hydrosols is less impacted by the tested temperatures in synthesis processes, while the calculated crystal size of TiO<sub>2</sub> is decreased by 21.0% if the synthetic temperature reduces from 50 °C to 40 °C. Moreover, compared with the crystalline size of TiO<sub>2</sub> particles in P25 nanopowder (50 to 100 nm), the crystalline sizes of TiO<sub>2</sub> particles in synthetic hydrosols are much smaller. Generally, for nano photocatalysts, a smaller crystalline size significantly influences the photoactivity. In coatings, small particle size is expected to cause a larger surface area to volume ratio, leading to better dispersion and higher reactivity [43,44]. The dispersion of synthetic TiO<sub>2</sub> hydrosols will be discussed in the following section.

# 3.2. Dispersion stability of TiO<sub>2</sub> hydrosols

For physically stable nanosuspensions, a minimum zeta potential of  $\pm$  30 mV is required in an electrostatic stabilization, and a minimum of  $\pm$  20 mV is sufficient for steric stabilization [45-47]. The hydrodynamic diameters and Zeta potential values of synthetic TiO<sub>2</sub> hydrosol samples are shown in Table 1. The hydrodynamic diameters (D<sub>H</sub>) of TiO<sub>2</sub> in hydrosol samples with the processing temperatures of 40 °C and 50 °C are 19 nm and 38 nm. The zeta potential of TiO<sub>2</sub> hydrosol synthesized at 40 °C and 50 °C are 43 mV and 39 mV, respectively. These results indicate that the synthesis temperature impacts the D<sub>H</sub> of TiO<sub>2</sub> in

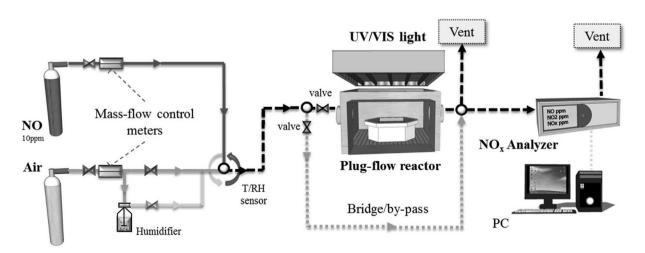


Fig. 1. The scheme of air purification experimental setup [6]

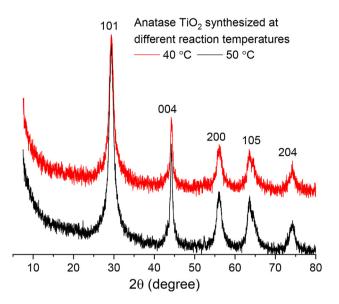


Fig. 2. XRD pattern of TiO<sub>2</sub> hydrosols dried at 105 °C.

hydrosol samples, while the zeta potential of TiO<sub>2</sub> hydrosols is less influenced. Compared with the primary crystalline sizes mentioned in Section 3.1, the D<sub>H</sub> of particles in TiO<sub>2</sub> hydrosol sample synthetic at 50 °C increases to 2.4 times of the primary particles, indicating the primary particles may have agglomerated because of the van der Waals forces. While the D<sub>H</sub> of particles in TiO<sub>2</sub> hydrosol sample synthetic at 40 °C increases to 1.6 times of the primary particles, demonstrating that the TiO<sub>2</sub> nanoparticles have relatively good dispersion and low agglomeration in this hydrosol form.

The polydispersity index (PI) is defined as the particle size distribution and its range of nanosuspensions. The physical stability of nanosuspensions can be evaluated by the PI value, indicating a nanosuspension with the smallest possible PI shows a long-time stability. A PI lower than 0.1 indicates a near-monodisperse particle distribution, a PI in the range of 0.1 to 0.25 shows a narrow size distribution of nanoparticles, and a PI larger than 0.5 means a very broad distribution of particle size [36,47,48]. As shown in Table 1, the PI values of the TiO<sub>2</sub> hydrosol synthesized at 40 °C and 50 °C are 0.11 and 0.14, respectively. Although the PI values in both TiO<sub>2</sub> hydrosol samples are larger than 0.1, it could be confirmed that the TiO<sub>2</sub> particles are closer to a near-monodisperse distribution in the hydrosol sample synthetic at 40 °C.

According to previous results [36], the mean particle size of P25 suspension ranges between 198 and 231 nm, with an average value of 221  $\pm$  11 nm, and an average PI ranges 0.2 to 0.6. Compared with the primary particles with the size of 30 to 50 nm, the TiO<sub>2</sub> particles in P25 suspension are agglomerates or aggregates with much larger particle sizes, and poor disperse stability. Compared with P25 suspension, these results prove that the TiO<sub>2</sub> hydrosol synthesized at 40 °C obtains smaller primary crystalline size, smaller hydrodynamic diameter and better disperse stability. Thus, the following characterization tests are focused on the comparison between TiO<sub>2</sub> hydrosol synthesized at 40 °C and P25 suspension.

#### 3.3. Optical analysis

The visual images and the UV–VIS absorbance spectra of  $TiO_2$ hydrosol and P25 suspension with the mass content are shown in Fig. 3. As can be seen from Fig. 3(a), TiO<sub>2</sub> hydrosol is transparent, while the P25 suspension sample with the same mass content is pure white and non-transparent. The transparent TiO<sub>2</sub> hydrosol is another reliable evidence proving that the TiO<sub>2</sub> particles in hydrosol are tiny and dispersed stable. The main reason for the white of P25 suspension is that the particle sizes of  $TiO_2$  aggregates in suspension are more significant than the wavelength of visible light, which reflects all wavelengths of visible light (400 to 800 nm) equally without wavelength selectivity. Moreover, as shown in Fig. 3(b), the absorbance of visible light in the range from 400 to 800 nm of P25 suspension is caused by the large aggregates of TiO<sub>2</sub> primary particles. The visual results confirm that the synthetic TiO<sub>2</sub> hydrosol presents good dispersion and better absorbance in the range of UV light.

# 3.4. BET specific surface area

The nitrogen isotherm plot of dried TiO<sub>2</sub> hydrosol is shown in Fig. 4 (a). The cumulative and differential pore size distribution plots of dried TiO<sub>2</sub> hydrosol are shown in Fig. 4 (b). In Fig. 4(a), the isotherm is a combination of type IV [49,50], which is associated with capillary condensation taking place in mesopores, and the limiting uptake over a range of high p/p0. The initial part of the Type IV isotherm, for example,  $0 < p/p_0 < 0.1$  in Fig. 4(a), is attributed to monolayer-multilayer adsorption since it followed the same path as the corresponding part of a Type II isotherm obtained with the given adsorptive on the same surface area of the adsorbent in a non-porous form. At a relative pressure between 0.4 and 0.8, there is one hysteresis loop, which corresponded to the smaller pore fraction and indicated that the pore size distributions of dried TiO<sub>2</sub> are unimodal pore size distributions in the mesoporous region [51]. Hysteresis appearing in the multilayer range of physisorption isotherms is usually associated with capillary condensation in mesopore structures. The shape of the hysteresis loop is type H2 associated, which could be observed in the pores with narrow necks and more extensive bodies (often referred to as 'ink bottle' pores) [50]. It could be seen from Fig. 4(b) that the dried TiO<sub>2</sub> hydrosol shows unimodal pore size distributions of small mesopores (about 3 nm). Due to the finer intra-aggregated pore, small mesopores are formed between TiO<sub>2</sub> grains or primary particles (represented by the hysteresis loop in the  $p/p_0$  range from 0.4 to 0.8).

Table 2 shows the BET specific surface area and pore parameters of dried TiO<sub>2</sub> hydrosol. Total pore volume is taken from the volume of nitrogen adsorbed at p/p0 = 0.985. The average pore diameter is estimated using the adsorption branch of the isotherm and the Barrett–Joyner–Halenda (BJH) formula. As shown in Table 2, the BET specific surface area of dried TiO<sub>2</sub> hydrosol is 244.8 m<sup>2</sup>/g, which is about five times larger than that of P25 (about 50 m<sup>2</sup>/g [3,34,52]). The absorption average pore width in dried hydrosol samples is 5.2 nm, indicating the mesopores have mainly existed among TiO<sub>2</sub> aggregates. The BET test results illustrate that the dried TiO<sub>2</sub> aggregates transformed from the hydrosol form contain

 Table 1

 Parameters of dispersion stability for synthetic TiO<sub>2</sub> hydrosols.

Synthesis temperatures	Hydrodynamic diameters± SD (d. nm)	Zeta potential(mV)	PI	Conductivity(mS/cm)
50 ℃	38 ± 16	39	0.14	0.12
40 ℃	19 ± 6	43	0.11	0.11

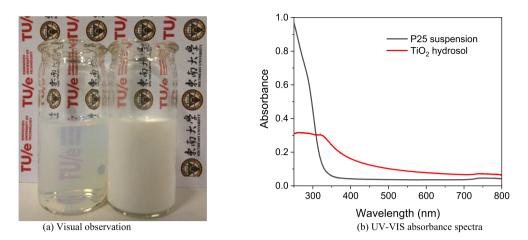


Fig. 3. The visual observation and UV–VIS absorbance spectra of TiO<sub>2</sub> hydrosol and P25 suspension (The left image in (a) is TiO<sub>2</sub> hydrosol, the right image in (b) is P25 suspension with the same dosage of 1.54 wt%).

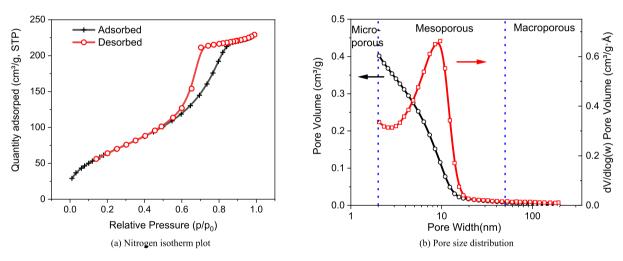


Fig. 4. The corresponding nitrogen isotherm plot and pore size distribution of dried TiO<sub>2</sub> hydrosol.

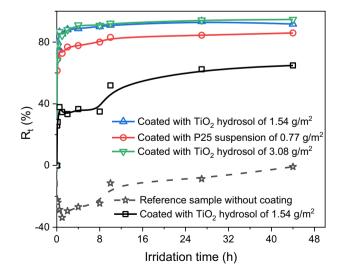
# Table 2 BET specific surface area and pore parameters of dried ${\rm TiO}_2$ hydrosol.

Hydrolysis	BET specific surface	Pore volume	Pore size
temperature (°C)	area (m²/g)	(ml/g)	(nm)
40	244.8	0.35	5.2

lots of mesopores and a relatively larger specific surface area, which are necessary for a coating with high photoactivity. The self-cleaning and air purification properties of dried coatings should be evaluated when applying photocatalytic coatings on the cement-based materials. The photocatalytic self-cleaning and air purification performances of  $TiO_2$  hydrosol-based coating on the surface of hardened cement paste will be discussed in Sections 3.5 and 3.6.

# 3.5. Self-cleaning performance

The uncoated hardened cement paste (HCP) sample and the HCP sample coated with the P25 suspension with the TiO<sub>2</sub> dosage of 1.54 g/m<sup>2</sup> are designed as the control samples in the self-cleaning tests. The colour change rate ( $R_t$ ) for RB of coated cement paste samples at the age of 7 days is shown in Fig. 5. The  $R_t$  values of HCP samples coated with TiO<sub>2</sub> hydrosol with the dosages of 0.77 g/m<sup>2</sup>, 1.54 g/m<sup>2</sup>, and 3.08 g/m<sup>2</sup> after four hours irradiation



**Fig. 5.** Percentage of colour change for RB on the surface of cement paste samples coated with different coatings after seven days hydration.

are 79.9%, 90.1%, and 91.1%, respectively. The  $R_t$  values of HCP samples coated with TiO<sub>2</sub> hydrosol with the dosages of 0.77 g/m<sup>2</sup>, 1.54 g/m<sup>2</sup>, and 3.08 g/m<sup>2</sup> after 24 h irradiation are 85.9%, 91.7%,

and 94.7%, respectively. The colour change of the blank cement sample is around zero after 24 h irradiation, indicating the blank cement presents very poor self-cleaning performance. The  $R_t$  of cement paste coated by P25 with the dosage of 1.54 g/m<sup>2</sup> after 24 h irradiation is only 64.9%. Moreover, in the TiO<sub>2</sub> hydrosol coated HCP samples, the  $R_t$  trends stabilize after two hours irradiation, while it needs at least 6 h irradiation to obtain a stable  $R_t$  in the P25 coated HCP sample. These results suggest that the photocatalytic self-cleaning performance of HCP paste coated with 1.54 g/m<sup>2</sup> TiO<sub>2</sub> in the form of hydrosol is much better than that of the form of P25 suspension.

According to the test results in the previous sections, the welldispersed  $TiO_2$  nanoparticles in dried  $TiO_2$  hydrosol coating contain lots of mesopores and obtain a large BET specific surface area, providing more reactive regions for the photochemical reactions between  $TiO_2$  and RB molecules. As a result, the  $TiO_2$  hydrosol coating shows excellent photoactivity in degrading the dried RB film on the surface of hardened cement paste.

# 3.6. Air purification performance

The air purification properties of the  $TiO_2$  hydrosol coating and P25 suspension coating are evaluated by the conversion rates of NO and NO<sub>x</sub> under UV irradiation. Fig. 6 shows the recorded degradation results of NO, NO<sub>2</sub>, and NO<sub>x</sub> of HPC panel samples coated with different dosages at the age of seven days. The calculated conversion rate of NO and NO<sub>x</sub> are shown in Table 3.

As shown in Fig. 6, the NO and  $NO_x$  conversion rates of the  $TiO_2$  hydrosol coating are barely influenced by the coated dosage of  $TiO_2$  nanoparticles, but that of the P25 suspension coating is impacted obviously. The NO concentration reduced from 1 ppm to about 0.05 ppm in a few minutes in the tested panel coated by  $TiO_2$ 

#### Table 3

The conversion rates of NO and  $\ensuremath{\mathsf{NO}}_x$  of two coatings on the surface of HPC panel samples.

Coated dosage (g/m <sup>2</sup> )	Coatings	NO conversion rate (%)	NO <sub>x</sub> conversion rate (%)
1.54	TiO <sub>2</sub> hydrosol	97.4	93.9
1.54	P25 suspension	97.7	95.0
3.08	TiO <sub>2</sub> hydrosol	98.9	96.1
3.08	P25 suspension	87.9	86.0

hydrosol with a coating dosage of  $1.54 \text{ g/m}^2$ . As to the panel sample coated by the TiO<sub>2</sub> hydrosol coating with the dosage of  $3.08 \text{ g/m}^2$ , the NO concentration decreases to about 0.1 ppm after a few minutes UV irradiation, and then continuously decreases to about 0.01 ppm after 120 min of UV irradiation. When coated with  $1.54 \text{ g/m}^2$  of P25 suspension, the NO concentration reduces to about 0.05 ppm from 1 ppm. However, when the coated dosage of P25 suspension increases to  $3.08 \text{ g/m}^2$ , the NO concentration reduces to about 0.15 ppm from 1 ppm. The coated dosage of the P25 suspension hampers the NO degrading efficiency, indicating the reactive region of TiO<sub>2</sub> particles reduces with the increase of P25 suspension coating.

As shown in Table 3, at a lower coated dosage of TiO<sub>2</sub> nanoparticles (1.54 g/m<sup>2</sup>), both NO and NO<sub>x</sub> conversion rates of two coatings are higher than 90%, and the NO and NO<sub>x</sub> conversion rates of the TiO<sub>2</sub> hydrosol and the P25 suspension are very close. When the coated dosage of TiO<sub>2</sub> nanoparticles increases to 3.08 g/m<sup>2</sup>, the NO and NO<sub>x</sub> conversion rates of the TiO<sub>2</sub> hydrosol coating are 1.5% and 2.3% higher than that with the coated dosage of 1.54 g/m<sup>2</sup>. However, with a higher coated dosage, the NO and NO<sub>x</sub> conversion rates of P25 suspension coating are 10% and 9.5% lower than those with a coated dosage of 1.54 g/m<sup>2</sup>.

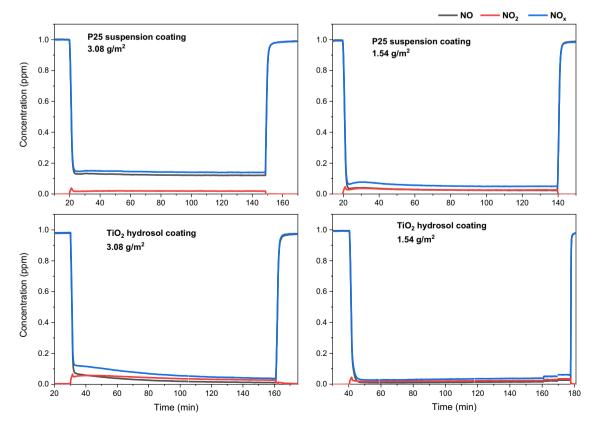


Fig. 6. NO, NO<sub>2</sub> and NO<sub>x</sub> concentration graphs of cement panel samples coated with TiO<sub>2</sub> hydrosol and P25 suspension coatings.

Here, comparing the degradation of organic dye film and inorganic gas pollutants, it is easy to find out that the synthetic TiO<sub>2</sub> hydrosol can degrade both RB and NO with high efficiencies. However, the P25 suspension shows unsatisfied efficiency in degrading RB. The reason for this significant difference may be related to the attachment between two kinds of target molecules and coatings. The RB water solution on the surface of coated hardened cement paste samples, after the evaporation of water, the RB molecules are distributed on the surface of coatings. While, NO and air fill the whole reactor evenly during the air purification test. Due to the small primary crystalline size, nano-scale hydrodynamic diameter and good disperse stability, the TiO<sub>2</sub> particles can be well dispersed in the dried TiO<sub>2</sub> hydrosol coating on the surface of hardened cement paste samples without too much agglomeration. The excellent dispersion means that the TiO<sub>2</sub> particles in dried hydrosol coating can provide more reactive areas for photodegrading RB and NO molecules. As a result, the synthetic TiO<sub>2</sub> hydrosol coating can obtain excellent photocatalytic self-cleaning performance and good air purification performance.

# 4. Conclusions

In this work, the nano dispersed  $TiO_2$  hydrosol is synthesized by an easy and low energy consumption method. The particle size and zeta potential are tested to evaluate the water disperse stability of  $TiO_2$  nanoparticles in hydrosol form. The specific surface area and the pore size distribution of dried  $TiO_2$  agglomerates from hydrosol form are measured by BET tests. The photocatalytic self-cleaning performance and air purification performance of  $TiO_2$  hydrosol by coating on the surface of hardened cement paste (HCP) samples are compared with that of commercial P25 nano  $TiO_2$  powder.

- (1) The primary crystalline size and hydrodynamic diameters of TiO<sub>2</sub> particles in hydrosol are influenced by the peptizing temperature. Compared with the primary particles, the TiO<sub>2</sub> nanoparticles in hydrosol with the synthetic temperature of 40 °C shows good dispersion and low agglomeration. The TiO<sub>2</sub> particles resulted from the dried synthetic hydrosol are pure anatase with a large BET specific surface area.
- (2) The TiO<sub>2</sub> hydrosol coatings with three different TiO<sub>2</sub> dosages present much better photocatalytic self-cleaning performance in degrading RB film on the surface of HCP samples. The degradation rate of RB increases with the coated dosage of TiO<sub>2</sub> particles in hydrosol coatings. Even with the lowest coated dosage (0.77 g/m<sup>2</sup>), more than 85% of RB molecules are degraded by the TiO<sub>2</sub> hydrosol coatings after 24 h of UV irradiation, while only 64.9% of RB molecules are degraded by the P25 suspension coating with the coated dosage of 1.54 g/m<sup>2</sup>.
- (3) The TiO<sub>2</sub> hydrosol coatings with different coated dosages present excellent air purification performance with the conversion rates of NO and NO<sub>x</sub> higher than 90%. Nevertheless, the higher the coated dosage of P25 suspension coating, the lower the conversion rates of NO and NO<sub>x</sub>. With a lower coated dosage ( $1.54 \text{ g/m}^2$ ), the P25 suspension coating presents the same good NO and NO<sub>x</sub> conversion ability with the TiO<sub>2</sub> hydrosol coating. While, with a higher coated dosage ( $3.08 \text{ g/m}^2$ ), the NO and NO<sub>x</sub> conversion rates of the P25 suspension coating reduce by around 10% compared with the rates with a lower coated dosage; on the contrary, the NO and NO<sub>x</sub> conversion rates of the TiO<sub>2</sub> hydrosol coating increase by about 2%.
- (4) When applying the nano TiO<sub>2</sub>-based coatings on cementbased materials, the high NO<sub>x</sub> conversion rates not always mean the excellent degradation efficiencies of organic dye.

Thus, evaluating the photocatalytic ability of coatings by degrading dried organic dyes is necessary for applying on cement-based materials.

#### **CRediT authorship contribution statement**

Zixiao Wang: Conceptualization, Methodology, Investigation, Data curation, Formal analysis, Validation, Writing - original draft. Florent Gauvin: Writing - review & editing. Pan Feng: Resources. H.J.H. Brouwers: Supervision, Writing - review & editing. Qingliang Yu: Supervision, Writing - review & editing.

#### **Declaration of Competing Interest**

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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